

Research Paper

Mechanical Properties and Porosity of Acrylic Cement Bone Loaded with Alendronate Powder

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Abstract

Aseptic loosening is the most common complication of joint replacement. Previous studies showed that acrylic bone cement loaded with a commercially-available alendronate powder (APAC) had good promise against wear debris-mediated osteolysis for prevention of aseptic loosening. The purpose of the present study was to investigate the effect of adding alendronate powder to an acrylic bone cement on quasi-static mechanical properties (namely, compressive strength, compressive modulus, tensile strength, and flexural strength), fatigue life, porosity, and microstructure of the cement. The results showed that adding up to 1 wt./wt.% alendronate powder exerted no detrimental effect on any of the quasi-static mechanical properties. However, the fatigue life of APAC decreased by between ~17% and ~27 % and its porosity increased by between ~ 5-7 times compared with corresponding values for the control cement (no alendronate powder added). Fatigue life was negatively and significantly correlated with porosity. Considering that fatigue life of the cement plays a significant role in joint replacement survival, clinical use of APAC cannot be recommended.

Key words: Acrylic bone cement; Alendronate; Mechanical properties; Fatigue life; Porosity

Introduction

Aseptic loosening is the most common complication of joint replacement [1]. Currently, there is no other treatment of aseptic loosening except revision arthroplasty [2]. The effective prevention is critical in the control of aseptic loosening. It has been shown that oral-administered bisphosphonate, especially alendronate, effectively inhibits wear debris-induced osteolysis and reduces the risk of aseptic loosening [3]. However, adverse events, notably, upper gastrointestinal ulcer, and a low bioavailability of 0.7% were frequently observed with oral-administered alendronate [4]. Furthermore, poor drug compliance in patients compromised the therapeutic efficacy of alendronate [5].

In 1994, Hamish *et al.* put forward the idea of a bisphosphonate-loaded acrylic bone cement as a local drug delivery system [6]. This proposal is similar to the addition of antibiotic(s) to an acrylic bone cement

[7]. The local high administration of bisphosphonate is achieved through sustained drug elution from the cement [8, 9]. Our previous *in vivo* study showed that the local release of alendronate from an acrylic bone cement inhibited wear debris-mediated osteolysis and enhanced the mechanical strength of the host bone in a cemented arthroplasty [10]. The proposed mechanism was that the eluted bisphosphonate stimulated an increase in osteoprotegerin ligand and a decrease in receptor activator of nuclear factor kappa-B ligand [11].

There have been a number of reports on bisphosphonate-loaded acrylic bone cement [12-15]. Yu *et al.* [12] found loading liquid pamidronate reduced the bending strength of the cement by 39%. Lewis *et al.* [13] demonstrated that the fatigue life of the cured cement was reduced by 97% and 39% when alendronate was added in liquid form and in the form

of tablet-ground powder, respectively (both values relative to that for the plain cement). These results were attributed to the interference of the polymerization process by the liquid alendronate and the excipient in the alendronate tablet [12, 13]. Matuszewski et al. [14] found that addition of 60 mg of pamidronate powder to 40 g of cement did not have a detrimental effect on either the compressive strength or the flexural strength of the cement. Song et al. [15] showed that addition of no more than 500 mg of alendronate powder to 50 g of cement did not significantly decrease either the compressive strength or the bending strength of the cement. To the best of our knowledge, there are no studies that involve determination of both quasi-static mechanical properties and fatigue life of an acrylic bone cement loaded with the powder of a bisphosphonate.

The purpose of the present study was to determine the effect of adding a commercially-available alendronate powder to an acrylic bone cement (APAC) on the compressive strength, tensile strength, flexural strength, fatigue life, porosity, and microstructure of the cement.

Materials and Methods

Materials and preparation of test specimens

The alendronate powder (Merck, Sharp & Dohme, Hangzhou, China; particle size range of 4.67 – 44.21 μm and mean = 20.77 μm) and the powder of the

bone cement (CMW Endurance; DePuy, London, England) were mixed using the open-bowl technique in ambient laboratory conditions, as described by Lewis et al. [13]. The liquid of the cement was added to the blended powder mixture. After that, the cement dough was injected into different stainless steel molds to produce specimens for different tests (Fig. 1). Then, the cured cement specimens were taken out of the molds and polished with a 600-grit sandpaper. The specimens were examined by visual inspection, digital calipering, high-light exposure, and X-radiography. Specimens were discarded if any of the following problems existed: surface flaw(s) > 0.25 mm on the body of the specimen and/or in transition sections; pore size > 1 mm in diameter within the specimen and/or in transition sections; and any dimension being more than 5% within its nominal value [13].

Study groups

The accepted specimens were divided randomly into eight groups based on the alendronate loading [6] and the aging condition of the cured cement (in ambient laboratory atmosphere or immersed in the phosphate buffered saline solution (PBS), pH 7.4 (Gibco Invitrogen Corp, Grand Island, NY) for 30 days) [16]. Thus, the groups were: the atmospheric control group (G_{A0}), the immersed control group (G_{I0}), the atmospheric 0.1 wt./wt.-alendronate group ($G_{A0.1}$), the immersed 0.1 wt./wt.-alendronate group ($G_{I0.1}$), the atmospheric 0.5 wt./wt. %alendronate group ($G_{A0.5}$), the immersed 0.1 wt./wt.-alendronate group ($G_{I0.5}$), the atmospheric 1 wt./wt.-alendronate group (G_{A1}), and the immersed 1 wt./wt.-alendronate group (G_{I1}).

Mechanical tests

The compressive strength, compressive modulus, tensile strength, and flexural strength were determined from tests carried out in ambient laboratory conditions (air, at 22 ± 1 °C), in accordance with ASTM F451 [17] ($n = 6-7$), ISO 527 [18] ($n = 6-8$), and ISO 5833 [19] ($n = 6-9$), respectively. The fatigue tests were conducted in PBS, at 37 ± 2 °C [13], in accordance with ASTM F2118 [20], with applied load corresponding to stress of ± 15 MPa and frequency of 5 Hz ($n =$

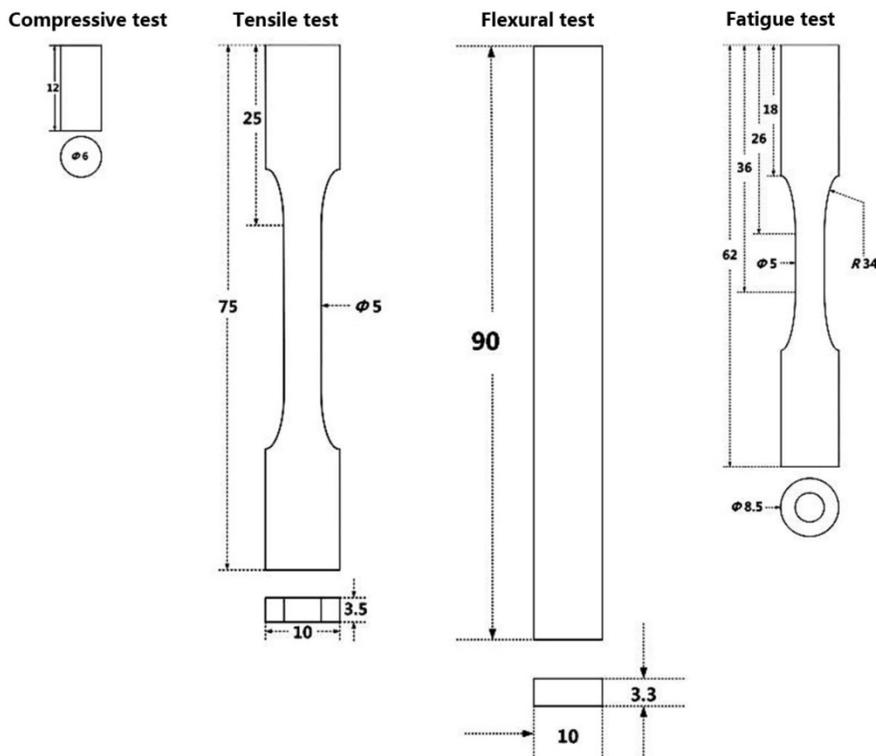


Figure 1. Dimensioned drawings of the specimens used in the mechanical properties tests (All dimensions are in mm).

7). The results of the fatigue tests (number of cycles to fracture (N_f)) were analyzed using the three-parameter Weibull relationship [21-23], leading to values of the minimum or guaranteed fatigue life (N_o), the characteristic fatigue life (N_a), and the Weibull modulus (m) [21-23]. The fatigue performance index was computed as the Weibull median (N_{WM}) thus [21]

$$N_{WM} = N_o + N_a(\ln 2)^{1/m}$$

Morphology and porosity of fracture surfaces

Following a fatigue test, the fracture surface of the specimen was cleaned ultrasonically, dried, sputter-coated with a gold-palladium layer, and inspected at 25.0 kV with a scanning electron microscope (SEM; SIRION 100, FEI, United States) at a magnification of 93x. Five sections of a specimen were selected at random and, then, porosity was determined. This involved using a commercially-available software package (Image-pro plus 6.0, Media Cybernetics, Inc., Silver Spring, MD, USA) to select a region of interest (ROI) within a segment and, then, digitizing it. For each image, the number of pores, the mean pore size, and the number of pixels counted were determined. Porosity (P) was calculated thus [10]

$$P = A_p/A_i,$$

where A_p is the total area occupied by all the pores and A_i is the area of the whole image.

Statistical analysis

Quantitative results were expressed as mean ± standard deviation. Each set of quantitative results was first tested for normality using the variance homogeneity and normal distribution tests. When this was found to be the case, a parametric test of significance of difference in means (one-way analysis of variance) was used. Otherwise, a non-parametric test of significance was used. Intergroup differences were compared with the method of the least significant difference. Regression analyses were performed between P and each of the quasi-static mechanical properties as well as between P and N_f [24]. Statistical significance was denoted if $p < 0.05$. Analyses were performed with a commercially-available software package (SPSS 16.0 for Windows; SPSS Inc., Chicago, IL, USA).

Results

Quasi-static mechanical properties

The influence of alendronate powder loading (ALN) on compressive strength was not significant ($F = 4.177, p = 0.094$), but immersion in PBS produced a

significant reduction in this strength ($F = 7.140, p = 0.0100$), and, for each cement group, this strength exceeded the minimum level stipulated in ASTM F451 (70 MPa) (Fig. 2). The same trends were found for (a) compressive modulus in terms of influence of ALN ($F = 4.306, p = 0.090$) and of immersion in PBS ($F = 15.833, p < 0.001$) (Fig. 2); and (b) tensile strength in terms of influence of ALN ($F = 6.732, p = 0.126$) and of immersion in PBS ($F = 23.125, p < 0.001$) (Fig. 2). Neither ALN nor immersion in PBS exerted a significant influence on flexural strength ($F = 0.005, p = 0.999$ and $F = 15.833, p < 0.001$, respectively) (Fig. 2). Furthermore, for each cement group, flexural strength was higher than the minimum level stipulated in ISO 5833 (50 MPa) (Fig. 2).

Fatigue life

Increase in ALN significantly decreased N_f and N_{WM} ($F = 4.306, p = 0.0090$) as did immersion in PBS ($F = 11.642, p = 0.01$) (Table 1).

Table 1. Summary of the fatigue test results (N_f) and the calculated Weibull parameters

Alendronate powder loading	Aging condition	N_f (cycles)	Weibull Parameters			
			N_o (cycles)	N_a (cycles)	m	N_{WM} (cycles)
0 wt./wt. %	Atmosphere	17724±5815	10404	19232	0.74	22136
	PBS	13458±2713 ^a	9414	15301	0.72	18595
0.1 wt./wt. %	Atmosphere	13701±2496 ^a	8103	15550	0.46	15101
	PBS	10875±1923 ^a	8518	14472	0.97	18504
0.5 wt./wt. %	Atmosphere	12973±2964 ^a	8955	14309	1.09	19114
	PBS	10980±1767 ^a	8690	11127	0.76	17562
1.0 wt./wt. %	Atmosphere	13398±2973 ^a	8604	11300	0.87	16062
	PBS	11291±1865 ^a	8349	11771	0.93	16236

^a Compared with the atmospheric control group, $p < 0.05$.

Porosity measures and pore morphology

While ALN significantly increased pore number for any pore size ($F = 43.463, p < 0.001$), the influence of immersion in PBS on a given porosity parameter was not significant ($F = 0.096, p = 0.759$) (Fig. 3, Table 2, and Fig. 4). Furthermore, for each cement group, the pores were uniformly distributed from the margin area to the core area of a specimen (Fig. 5).

Table 2. Microstructure parameters of fracture surface of specimens

Alendronate powder loading (wt./wt. %)	Aging condition	Number of pores	Total pore area (mm ²)	Porosity (%)
0	Atmosphere	3.33±2.42	0.051±0.030	0.90±0.71
	PBS	2.86±1.46	0.062±0.030	0.90±0.47
0.1	Atmosphere	87.29±44.77 ^a	0.017±0.010 ^a	5.33±1.56 ^a
	PBS	79.29±26.76 ^a	0.016±0.004 ^a	5.17±1.12 ^a
0.5	Atmosphere	75.71±33.08 ^a	0.019±0.007 ^a	6.03±1.69 ^a
	PBS	88.57±25.75 ^a	0.014±0.004 ^a	5.27±0.58 ^a
1	Atmosphere	84.14±31.18 ^a	0.018±0.007 ^a	5.93±1.28 ^a
	PBS	103.29±59.13 ^a	0.017±0.011 ^a	6.36±2.45 ^a

^a Compared with the atmospheric control group or immersed control group, $p < 0.001$.

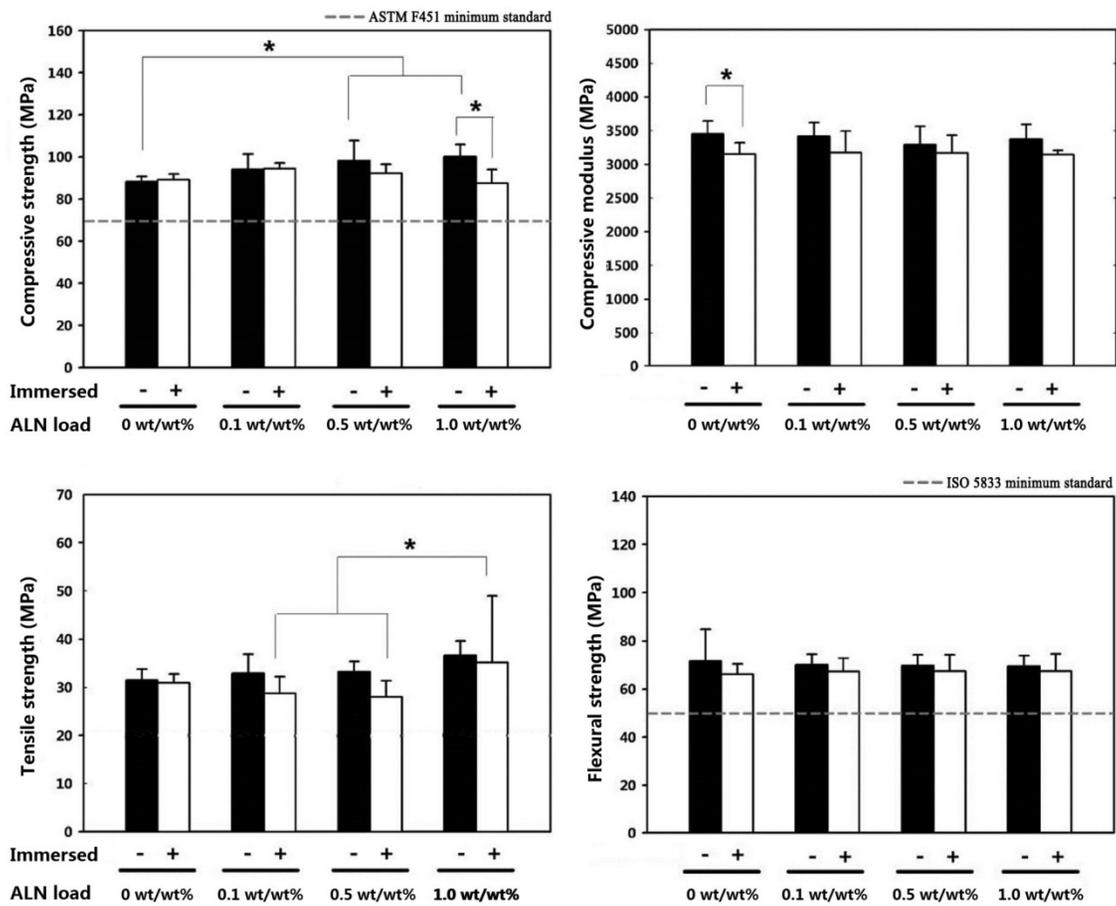


Figure 2. Summary of the quasi-static mechanical properties.

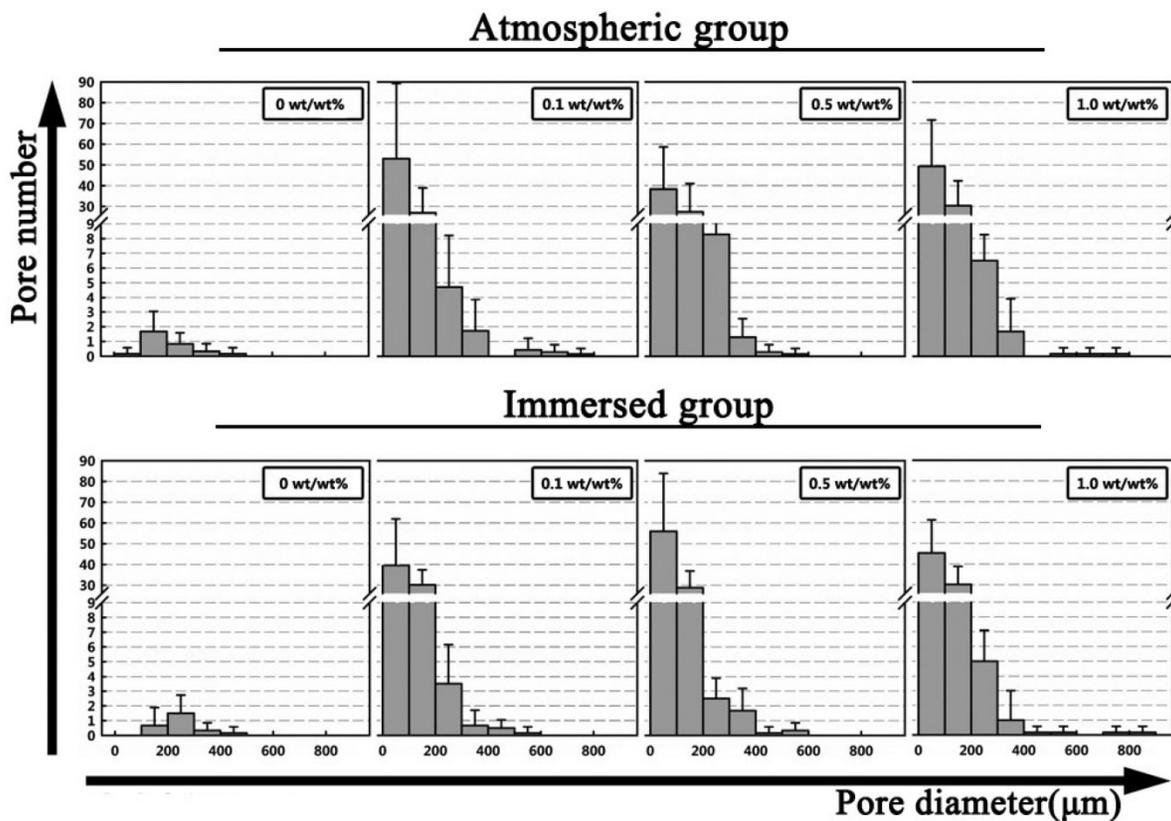


Figure 3. Summary of porosity measures.

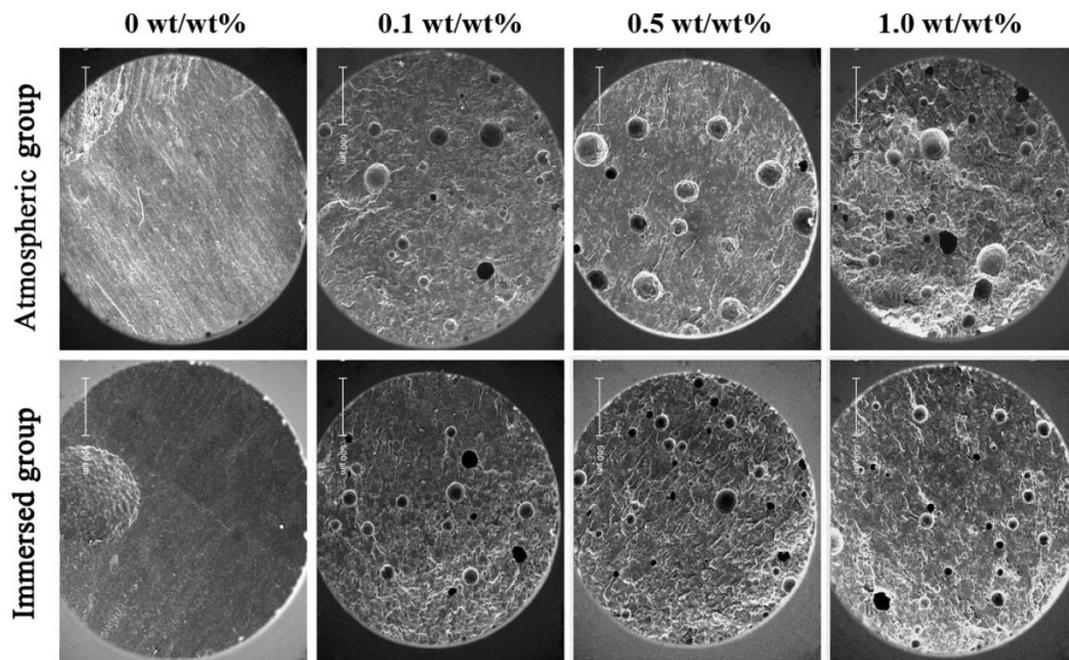


Figure 4. The micrographs of the surfaces of cement specimens after fracture in the fatigue test

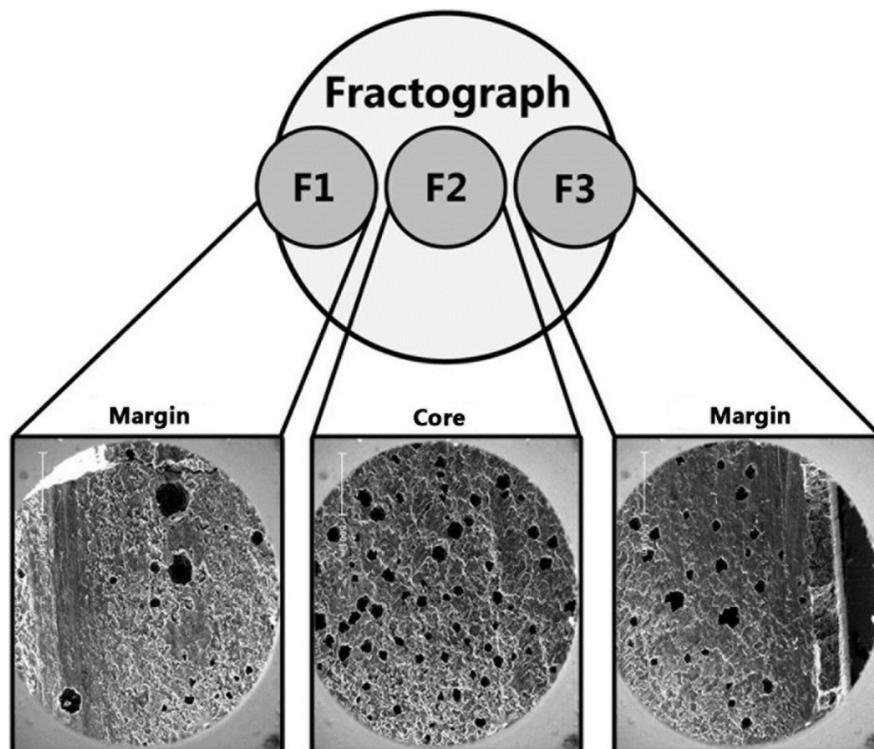


Figure 5. A sample of continuous scanning images of the surfaces of cement specimens after fracture in the fatigue test (G_{A1} cement group specimens).

SEM images showed that G_{A0} and G_{I0} specimens had few pores while specimens in each of the other six alendronate groups had a large number of small particles ranging in size from $10\ \mu\text{m}$ to $200\ \mu\text{m}$ (Fig. 4). The pores were distributed from the marginal area to the core area of the specimen (G_{A1} as an example in Fig. 5).

Correlations

While the correlation between P and each of the quasi-static mechanical properties was weak and not statistically significant ($p > 0.05$) (Fig. 6), porosity had a negative and significant correlation with fatigue life ($R^2 = 0.562$, $p < 0.01$) (Fig. 7).

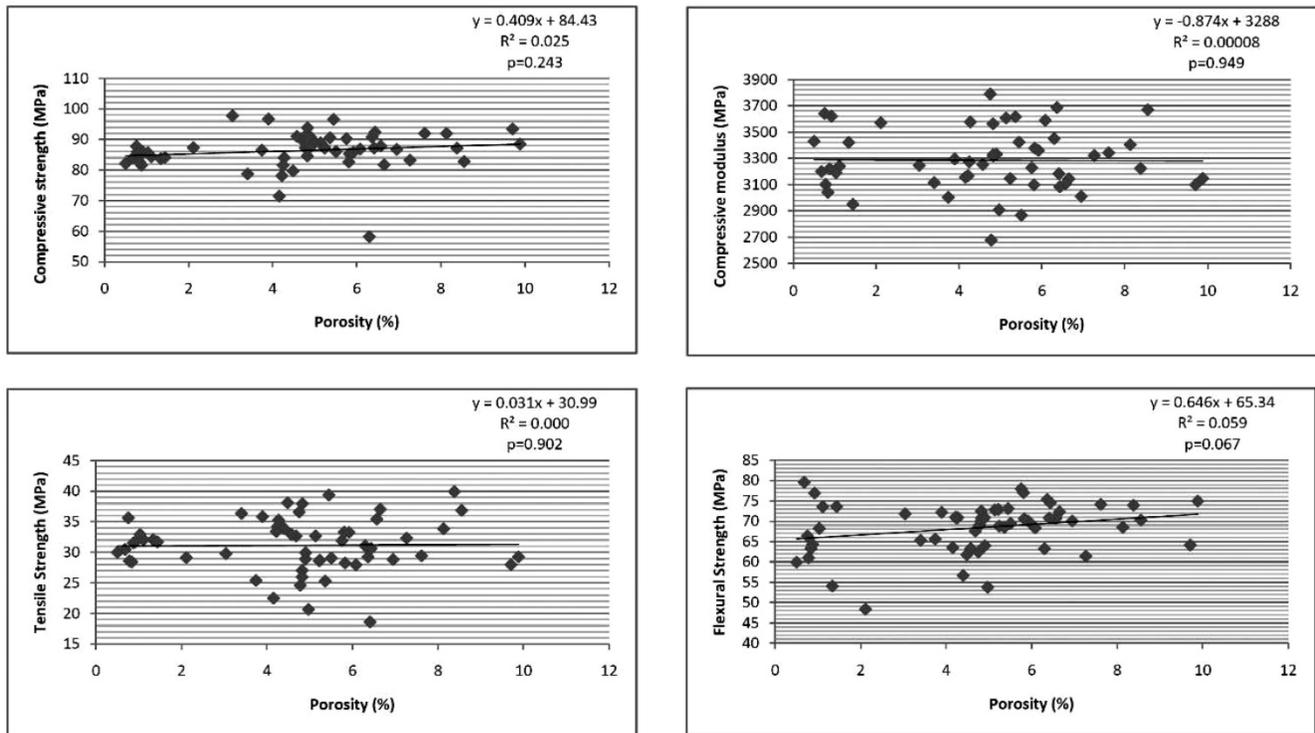


Figure 6. Correlations between porosity and each of the quasi-static mechanical properties.

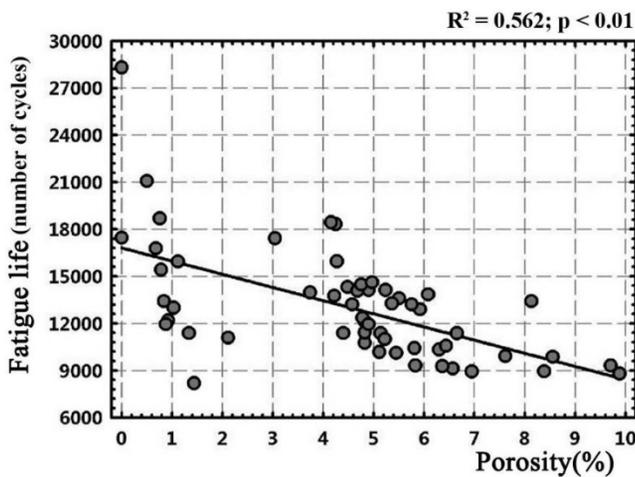


Figure 7. Correlation between porosity and fatigue life (N_f).

Discussion

The mechanical strength of acrylic cement is highly related with the lifespan of the cemented arthroplasty [25]. In previous studies, it was showed that APAC could locally release bisphosphonate, inhibit periprosthetic bone resorption, and enhance the mechanical strength of host bone while avoiding the detrimental effect of oral medication [8-11]. Incorporation of liquid bisphosphonate or tablet-ground alendronate powder significantly decreased the mechanical strength of acrylic bone cement [12, 13]. APAC may be a better option to maintain the mechanical strength of cement [14, 15].

In the present study, we evaluated quasi-static mechanical properties, fatigue life, porosity, and microstructure of APAC.

Our microscopic results demonstrated increased porosity of APAC (relative to that of the control cement) and propose two possible mechanisms for this finding. First, previous studies found that, just like as in the case of an antibiotic-loaded acrylic bone cement, elution of a bisphosphonate powder from a bone cement is low [8]. Since, in the case of APAC, the pores were distributed throughout the fracture surface, we assumed that the large number of pores could not result from powder aggregation. Second, given that the control group and the APAC group specimens were all cured under the same conditions, air inclusion could be eliminated as the reason.

Crack initiation, propagation, and accumulation is the cause of cement fracture. It results in mechanical failure of a cemented arthroplasty [26]. Topoleski et al. [26] and Hoey et al. [24] proposed that, in an acrylic bone cement, pores play a dual role: on one hand, they cause stress concentration and promote crack initiation and propagation, which is detrimental for the cement mantle; but, on the other hand, they divert energy to the surroundings and prevent crack propagation, which is beneficial for the cement mantle. In the present study, a strong and significantly negative correlation was found between fatigue life and porosity, but the correlation between each of the quasi-static mechanical properties and

porosity was weak and not statistically significant. These findings suggest that in the fatigue test, pores promoted cracks instead of inhibition, which agreed with the conclusions of previous studies [24, 27], but, in a quasi-static test, the effect of crack promotion is offset by that of crack inhibition.

In previous studies, it was shown that liquid bisphosphonate significantly reduced the mechanical strength of acrylic bone cement [12]. It was speculated that liquid bisphosphonate dilutes the substrate and the catalyst of polymerization [12]. The residual monomer acted as a stress riser, which led to mechanical failure. When tablet-ground alendronate powder was added to an acrylic bone cement, fatigue life was reduced by 39% [13]. The hand-made preparation of the powder and the mixed excipient in the tablet may account for this unsatisfactory result. In the present study, we used a commercially-available alendronate powder, which does not contain any excipient mixed. Nonetheless, the fatigue life of APAC decreased by a much smaller margin compared to the case when tablet-ground alendronate powder was used [13].

We immersed all the specimens in PBS, at 37 °C, for 30 days to simulate the physiological environment [16]. The mechanical strength after immersion changed due to two effects: the aging effect increased the strength in the early time and the plasticization effect of PBS uptake decreased the strength in the late time [16]. In our study, all the specimens were tested after 30 days' standing to mitigate the aging effect as far as possible. Quasi-static mechanical properties decreased after PBS immersion, suggesting that the plasticization effect might play an important role.

Three study limitations are recognized. First, porosity was determined using SEM. A 3D technique, such as micro-computed tomography, may provide a more reliable and precise evaluation of number, size, and distribution of pores in bone cement than SEM [28]. Second, the fatigue tests were conducted at only one stress level (± 15 MPa); as such, fatigue limits of the cements were not determined. Third, the specimens were prepared using hand-mixed, instead of vacuum mixed, cement [13]. It has been suggested that vacuum mixing decreases porosity and increases fatigue life of an acrylic bone cement [29]. However, Macaulay et al. [30] found that hand mixing by skilled technicians did not result in negative effects on cement porosity. Hence, hand mixing is thought to be reliable and reproducible [29, 30].

In conclusion, we found that adding a commercially-available alendronate powder to an acrylic bone cement (APAC) did not significantly affect any of the quasi-static mechanical properties of the cement determined but significantly decreased the

fatigue life of the cement, which may be a consequence of the significant increase in porosity. Thus, clinical use of APAC cannot be recommended. Future work will investigate reinforcement of APAC with additives, such as bioceramic fibers [31], as means of increasing the fatigue life of the cement.

Acknowledgments

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Abbreviations

APAC: acrylic bone cement loaded with alendronate powder

PBS: phosphate buffered saline

ASTM: American Society for Testing and Materials

ISO: International Standardization Organization

SEM: scanning electron microscope

Competing Interests

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The authors have declared that no competing interest exists.

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